# Crystal Structure and Metal Distribution of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ 

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Single crystals of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$, a mixed-valence compound of $\mathbf{V}^{\mathrm{IV} / \mathbf{v}}$, were hydrothermally synthesized. It crystallizes in the orthorhombic system Ibam with $a=14.3298(6) \AA, \quad b=$ $\operatorname{9.8906}(6) \AA, c=8.3950(8) \AA$, and $Z=8$. The structure was refined to $R=0.034$ and $R_{\mathrm{w}}=0.030$ for 1558 reflections with $I>3 \sigma(I)$. There are three kinds of metal sites, namely, octahed$\operatorname{ral} M(16 k)$ for $M=\operatorname{Co}, V(1)$, tetrahedral $\mathrm{V}(2)(8 j)$, and trig-onal-bipyramidal $V(3)(8 j)$, where the $M$ site was found to consist of half-occupied Co and $\mathrm{V}(1)$. Its polyhedral framework is built up of $M O_{6}$ slabs and $\mathrm{V}(2) \mathrm{O}_{4}-\mathrm{V}(3) \mathrm{O}_{5}$ units. The $M \mathrm{O}_{6}$ slab is constructed by linking zigzag chains of edge-sharing $\mathrm{MO}_{6}$ octahedra running along the $c$ axis. The linkage is made along the $b$ axis by sharing common vertices of $O(5)$ and thus the slab stands in the be plane. The $\mathrm{V}(2) \mathrm{O}_{4}-\mathrm{V}(3) \mathrm{O}_{5}$ unit, playing a role in bridging $M \mathrm{O}_{6}$ slabs, is made of an edge-sharing $\mathrm{V}(3) \mathrm{O}_{5}$ pair to which two $\mathrm{V}(2) \mathrm{O}_{4}$ are attached. Valence states of metals were evaluated as $\mathbf{C o}^{\text {II }}, \mathrm{V}(1)^{\text {IV }}, \mathrm{V}(2)^{\mathbf{V}}$, and $\mathrm{V}(3)^{\mathbf{V}}$. The metal distribution over the $M$ site is basically random but is constrained by a rule to form a $\mathrm{Co}-\mathrm{O}(5)-\mathrm{V}(1)$ group at the linkage of $\mathrm{MO}_{6}$ chains, which accounts for the half-occupancies of $\mathbf{C o}$ and $V(1)$ as well as a positional shift of $\mathbf{O ( 5 )}$. The structure is compared with those of $\alpha-\mathrm{ZnV}_{3} \mathrm{O}_{8}$ and $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$ in particular reference to metal distributions. The magnetic susceptibility curve of $\alpha$ $\mathrm{CoV}_{3} \mathrm{O}_{8}$ exhibits a sharp peak at 8 K , suggesting the onset of antiferromagnetic order. © 1998 Academic Press

## INTRODUCTION

There are various phases of mixed-valence vanadium (IV, V ) oxides in ternary systems $M-\mathrm{V}-\mathrm{O}$ with foreign $M$ meta1s. A major part of them, especially for $M=$ alkali or alka-line-earth metals, are known as the vanadium oxide bronzes (VOB), which have been attracting much attention because of their structural and physical properties (1). VOBs have general structures consisting of $\mathrm{V}-\mathrm{O}$ polyhedral frameworks and interstitial foreign metals. On the other hand, phases with $M=3 d$ transition metals are expected to exhibit framework structures made up of both $\mathrm{V}-\mathrm{O}$ and $M-\mathrm{O}$ polyhedra. However, only a few $M-\mathrm{V}-\mathrm{O}$ compounds of this class are known. For example, in the $\mathrm{CoO}-\mathrm{VO}_{2}-\mathrm{V}_{2} \mathrm{O}_{5}$
system there is only one definite mixed-valence compound, $\mathrm{CoV}_{3} \mathrm{O}_{8}$, originally formulated as $\mathrm{Co}_{1+y}\left(\mathrm{~V}_{3} \mathrm{O}_{8}\right)_{2}(0.90 \leq$ $y \leq 1)(2)$. This is also the case for the $\mathrm{NiO}-\mathrm{VO}_{2}-\mathrm{V}_{2} \mathrm{O}_{5}$ system (2). $\mathrm{CoV}_{3} \mathrm{O}_{8}$ was reported to adopt the $\alpha$ form below $650^{\circ} \mathrm{C}$ and transforms reversibly into a high-temperature phase of the $\beta$ form; the $\alpha$ form crystallizes in the bodycentered orthorhombic system Ic $2 a$ or Icma with cell parameters $a=14.234(10) \AA, b=8.390(8) \AA$, and $c=9.822(9) \AA$ and the $\beta$ form crystallizes in the $C$-centered monoclinic system with $a=10.297(9) \AA, b=8.524(7) \AA, c=7.774(9) \AA$, and $\beta=119.1(4)^{\circ}(2)$. Isomorphous phases were found for other divalent metals of $M=\mathrm{Ni}, \mathrm{Zn}$, and $\mathrm{Mg}: \mathrm{MgV}_{3} \mathrm{O}_{8}$ and $\mathrm{ZnV}_{3} \mathrm{O}_{8}$ exhibit both the $\alpha$ and $\beta$ forms $(2,3)$ whereas $\mathrm{NiV}_{3} \mathrm{O}_{8}$ exhibits the $\alpha$ form and an undefined $\beta$ form (2). Though the structures of $\alpha$ - and $\beta-\mathrm{CoV}_{3} \mathrm{O}_{8}$ remain unknown, those of $\alpha-\mathrm{ZnV}_{3} \mathrm{O}_{8}$ (4) and $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$ (5) have been determined and both phases adopt similar framework structures consisting of $\mathrm{MO}_{6}$ octahedra and $\mathrm{V}-\mathrm{O}$ polyhedra. Thus, in these mixed-valence $M \mathrm{~V}_{3} \mathrm{O}_{8}$ phases, in contrast to the VBO phases, both $M$ and V metals construct polyhedral frameworks. Further insight into the crystal chemistry of the $M \mathrm{~V}_{3} \mathrm{O}_{8}$ phases should be of interest.

In the present study single crystals of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ have been hydrothermally grown and its structure has been solved. The structure is closely related to that of $\alpha-\mathrm{ZnV}_{3} \mathrm{O}_{8}$ but has a different metal distribution as well as a different space group; this point is discussed together with the structure of $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$. The magnetic susceptibility measurement suggested antiferromagnetic order below 8.2 K.

## EXPERIMENTAL

## Sample Preparation

Starting materials for V and Co sources were $\mathrm{VO}(\mathrm{OH})_{2}$ powders and $\mathrm{CoI}_{2}$ or $\mathrm{CoCl}_{2}$ aqueous solutions. $\mathrm{VO}(\mathrm{OH})_{2}$ powders were prepared in advance by the hydrothermal treatment of a mixture of $\mathrm{VOSO}_{4}$ and NaOH solutions at $150^{\circ} \mathrm{C}$. A suspension of $\mathrm{VO}(\mathrm{OH})_{2}$ powders $(0.5 \mathrm{~g})$ in a $\mathrm{CoI}_{2}$ or $\mathrm{CoCl}_{2}$ solution ( $80 \mathrm{ml}, 0.1 \mathrm{moll}^{-1}$ ) was sealed in a Pyrex
ampoule and treated hydrothermally in an autoclave at $280^{\circ} \mathrm{C}$ for 48 h . Precipitates were separated by filtration and washed thoroughly with distilled water and ethanol. The products consisted of shiny black granules and black-brown powders. The black granules were easily separated by repeated decantation and sonication and appeared to be single-crystalline particles under an optical microscope. The crystalline phase was identified as $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ (2) by powder X-ray diffractometry. The $\mathrm{Co} / \mathrm{V}$ atomic ratio was obtained to be 0.333 (2) by energy-dispersive X-ray analysis. The byproduct of the black-brown powders was found to be the $\mathrm{VO}_{2}(A)$ phase (6). A magnetic susceptibility measurement was carried out on as-synthesized $\mathrm{CoV}_{3} \mathrm{O}_{8}$ granules by using a SQUID magnetometer from 5 to 300 K under a magnetic field of 1000 G .

## Single-Crystal X-Ray Diffraction Study

A single crystal of $\mathrm{CoV}_{3} \mathrm{O}_{8}$ with dimensions of $0.25 \times$ $0.15 \times 0.05 \mathrm{~mm}$ was mounted on a Rigaku AFC-7R diffractometer with monochromatized Mo $K \alpha$ radiation. The crystal system is orthorhombic with cell parameters $a=$ $14.3298(6) \AA, \quad b=9.8906(6) \AA$, and $c=8.3950(8) \AA$ determined from 25 reflections of $37.77^{\circ}<2 \theta<39.90^{\circ}$. A unit cell volume of $1189.8(1) \AA^{3}$ corresponds to $Z=8$. The systematic extinction gave possible space groups Ibam and Iba2. Data collection was performed up to $2 \theta=80^{\circ}$ using the $\omega-2 \theta$ scanning method $(\Delta \omega=1.42+0.30 \tan \theta)$ and no significant intensity decay ( $-0.8 \%$ ) was detected from standard reflections monitored every 150 reflections. A total of 2072 reflections were collected, of which 1558 reflections with $I>3 \sigma(I)$ were used in the structure refinements. Empirical absorption corrections based on $\psi$ scan were applied, resulting in transmission coefficients $0.580-1.000$. All calculations of data processing and structural analysis were performed using the TEXSAN crystallographic software package (7).
The structure was determined by the following procedure. The space group Ibam was chosen since the statistical treatment of intensity data strongly preferred centrosymmety; a comparison with the space group $I b a 2$ is discussed later. Three metal sites were located by the direct method, namely, one $16 k$ and two $8 j$ positions. Placing V atoms tentatively in the metal sites, we successfully located six oxygen sites in difference Fourier maps, namely, two $16 k$ positions $(\mathrm{O}(1)$, $\mathrm{O}(6))$, three $8 j$ positions $(\mathrm{O}(2), \mathrm{O}(3), \mathrm{O}(4))$, and one $8 f$ position $(\mathrm{O}(5))$. Oxygen coordinations of the metal sites were found to be octahedral ( 16 k ), tetragonal ( $8 j$ ), and trigonal bipyramidal (8j). The structure was successfully constructed by placing Co and $V(1)$ atoms independently in the octahedral position with half-occupancies and $\mathrm{V}(2)$ and $\mathrm{V}(3)$ atoms in the tetragonal and trigonal-bipyramidal positions, respectively. The occupancies of Co and $\mathrm{V}(1)$ sites were checked to show the same value of $0.498(3)$ and thus fixed to

TABLE 1
Crystallographic Data and Experimental Parameters for $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$

| Space group | Ibam |
| :--- | :---: |
| $a(\AA)$ | $14.3298(6)$ |
| $b(\AA)$ | $9.8906(6)$ |
| $c(\AA)$ | $8.3950(8)$ |
| $V\left(\AA^{3}\right)$ | $1189.8(1)$ |
| $Z$ | 8 |
| $D_{\mathrm{c}}\left(\mathrm{g} \mathrm{cm}^{-3}\right)$ | 3.794 |
| $\mu\left(\mathrm{~cm}^{-1}\right)$ | 71.70 |
| Number of reflections $(I>0)$ | 2072 |
| Number of reflections $(I>3 \sigma(I))$ | 1558 |
| Number of variables | 67 |
| $R$ | 0.034 |
| $R_{\mathrm{w}}$ | 0.030 |

0.5 . Co and $\mathrm{V}(1)$ atoms were refined isotropically because anisotropic $U_{22}$ values for both atoms became almost zero or even negative presumably due to strong correlation between their positional and thermal parameters. V(2) and $\mathrm{V}(3)$ sites were confirmed to be fully occupied. $\mathrm{O}(5)$ atom initially placed in the 8 f position exhibited large anisotropic displacements $\quad\left(U_{11}=0.015(1), \quad U_{22}=0.031(2), \quad U_{33}=\right.$ $\left.0.019(1), B_{\text {eq }}=1.72(5) \AA^{2}\right)$, suggesting an appropriate rearrangement of the $O(5)$ position. The $O(5)$ position was changed from $8 f$ to $16 k$ and from full occupancy to halfoccupancy by introducing parameters of $y=\delta_{1}$ and $z=\frac{1}{4}-\delta_{2}$, which were refined to $\delta_{1}=0.011(1)$ and $\delta_{2}=$ $0.013(2)$, and thereby the anisotropy was depressed $\left(U_{11}=\right.$ $\left.0.016(1), U_{22}=0.018(4), U_{33}=0.006(4), B_{\text {eq }}=1.1(1) \AA^{2}\right)$. The structural basis of this rearrangement is described later. The revised model was refined to $R=0.034$ and $R_{\mathrm{w}}=0.030$. The experimental and crystallographic data are listed in Table 1 and the atomic parameters and the isotropic temperature factors are in Table 2.

TABLE 2
Atomic Parameters, Isotropic Temperature Factors, and Occupancies for $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$

| Atom Position | $x$ | $y$ | $z$ | $B_{\text {eq }}\left(\AA^{2}\right)$ | Occupancy |  |
| :--- | :---: | :--- | :--- | :--- | :--- | :---: |
| Co | $16 k$ | $0.1577(1)$ | $0.8287(2)$ | $0.3089(2)$ | $0.58(3)$ | 1 |
| $\mathrm{~V}(1)$ | $16 k$ | $0.1521(1)$ | $0.8363(2)$ | $0.3112(3)$ | $0.62(3)$ | 1 |
| $\mathrm{~V}(2)$ | $8 j$ | $0.29849(4)$ | $0.94395(6)$ | 0 | $0.504(8)$ | 1 |
| $\mathrm{~V}(3)$ | $8 j$ | $0.02252(4)$ | $0.66658(6)$ | 0 | $0.614(8)$ | 1 |
| $\mathrm{O}(1)$ | $16 k$ | $0.0789(1)$ | $0.7242(2)$ | $0.1573(2)$ | $1.37(4)$ | 1 |
| $\mathrm{O}(2)$ | $8 j$ | $0.5824(2)$ | $0.2761(3)$ | 0 | $0.92(4)$ | 1 |
| $\mathrm{O}(3)$ | $8 j$ | $0.7337(2)$ | $0.4125(3)$ | 0 | $0.90(4)$ | 1 |
| $\mathrm{O}(4)$ | $8 j$ | $0.4200(2)$ | $0.9826(3)$ | 0 | $1.01(4)$ | 1 |
| $\mathrm{O}(5)$ | $16 k$ | $0.1086(2)$ | $0.011(1)$ | $0.237(2)$ | $1.1(1)$ | 1 |
| $\mathrm{O}(6)$ | $16 k$ | $0.2678(1)$ | $0.8531(2)$ | $0.1638(2)$ | $0.88(3)$ | 1 |

## RESULTS AND DISCUSSION

## Description of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ structure

Figure 1 depicts polyhedral representations of the $\alpha$ $\mathrm{CoV}_{3} \mathrm{O}_{8}$ structure consisting of three polyhedral types: $M \mathrm{O}_{6}(M=\mathrm{Co}, \mathrm{V}(1))$ octahedra, $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra, and $\mathrm{V}(3) \mathrm{O}_{5}$ trigonal bipyramids. Polyhedral units are presented in Fig. 2a for $\mathrm{MO}_{6}-M \mathrm{O}_{6}$ and Fig. 2b for $\mathrm{V}(2) \mathrm{O}_{4}-\mathrm{V}(3) \mathrm{O}_{5}$. Bond distances and angles for the polyhedra are listed in Table 3: as seen in Fig. 2a, two bond distances can be obtained for $\mathrm{Co}-\mathrm{O}(5)(1.78(1), 2.02(1) \AA$ ) and $\mathrm{V}(1)-\mathrm{O}(5)$ (1.69(1), 1.94(1) $\AA$ ), from which the longer one of $\mathrm{Co}-\mathrm{O}(5)$ and the shorter one of $\mathrm{V}(1)-\mathrm{O}(5)$ are adopted from the considerations described in the next section. The polyhedral framework is constructed in the following manner. $\mathrm{MO}_{6}$ octahedra are joined by sharing edges of $\mathrm{O}(2)-\mathrm{O}(3)$ and $\mathrm{O}(6)-\mathrm{O}(6)$ to form a zigzag chain running parallel to the $c$ axis. The chains are connected by sharing $\mathrm{O}(5)$ vertices along the $b$ axis to form an $M \mathrm{O}_{6}$ octahedral slab standing in the $b c$ plane as seen a corrugated wall in Figs. 1b and 3a. As depicted in Fig. 2b, a $\mathrm{V}(3) \mathrm{O}_{5}$ dimer is made by sharing an $\mathrm{O}(4)-\mathrm{O}(4)$ edge to which two $\mathrm{V}(2) \mathrm{O}_{4}$ are attached by sharing $\mathrm{O}(4)$ vertices from opposite sides to form a $\mathrm{V}(2) \mathrm{O}_{4}{ }^{-}$ $\mathrm{V}(3) \mathrm{O}_{5}$ unit. The $M \mathrm{O}_{6}$ slabs are bridged by the $\mathrm{V}(2) \mathrm{O}_{4}{ }^{-}$ $\mathrm{V}(3) \mathrm{O}_{5}$ units sharing $\mathrm{O}(3), \mathrm{O}(6)$, and $\mathrm{O}(6)$ vertices of $\mathrm{V}(2) \mathrm{O}_{4}$ and $\mathrm{O}(1), \mathrm{O}(1)$, and $\mathrm{O}(2)$ vertices of $\mathrm{V}(3) \mathrm{O}_{5}$. The framework structure exhibits an elongated hexagonal tunnel running parallel to the $c$ axis as seen in Fig. $1 b$.

## Valence States and Distribution of Co and V Atoms over the octahedral $M$ site

Bond valence sums (BVS) (8) for metals and oxygens are listed in Table 4. The results indicate that both $\mathrm{V}(2)$ and $\mathrm{V}(3)$ are obviously pentavalent and that Co and $\mathrm{V}(1)$ are probably di- and tetravalent, respectively. As mentioned before, metal distribution over the $M$ site can yield two values each for $\mathrm{Co}-\mathrm{O}(5)$ and $\mathrm{V}(1)-\mathrm{O}(5)$ distances that greatly affect the BVS values of $\mathrm{Co}, \mathrm{V}(1)$, and $\mathrm{O}(5)$. The best BVS values for $\mathrm{Co}^{\text {II }}, \mathrm{V}(1)^{\text {IV }}$, and $\mathrm{O}(5)^{\text {II }}$, listed in Table 4, were obtained from the bond distances given in Table 3. The metal distribution in the $M \mathrm{O}_{6}$ slab to give the best BVS values is illustrated in Fig. 3: Fig. 3a shows how the $M \mathrm{O}_{6}$ slab is built of $M \mathrm{O}_{6}$ chains by sharing $\mathrm{O}(5)$ vertices and Fig. 3 b shows the corresponding atomic arrangement. As shown in Fig. 3b and also in Fig. 2a, the Co and $\mathrm{V}(1)$ atoms occupy metal positions on opposite sides of the $\mathrm{O}(5)$ site and the $\mathrm{O}(5)$ atom occupies the site closer to the $\mathrm{V}(1)$ atom, resulting in the bond distances of $\mathrm{Co}-\mathrm{O}(5)=2.02 \AA$ and $\mathrm{V}(1)-\mathrm{O}(5)=$ $1.69 \AA$. However, the BVS values of 2.37 for $\mathrm{Co}^{\mathrm{II}}, 3.66$ for $\mathrm{V}(1)^{\mathrm{IV}}$, and 1.70 for $\mathrm{O}(5)^{\mathrm{II}}$ somewhat deviate from the ideal values. This is due to the difficulty in determining the exact positions of Co and $V(1)$, which overlap each other. To sum up, as seen in Fig. 3b, the metal distribution over the $M$ site


FIG. 1. Polyhedral representations of the crystal structure of $\alpha$ $\mathrm{CoV}_{3} \mathrm{O}_{8}$ : (a) perspective view; (b) projected view onto the $a b$ plane.
is random in the $M \mathrm{O}_{6}$ chain but constrained by a rule to give $\mathrm{Co}-\mathrm{O}(5)-\mathrm{V}(1)$ bonding between neighboring chains through the shared vertex $\mathrm{O}(5)$. This constraint results in the half-occupancies of Co and $\mathrm{V}(1)$ sites as well as the $\mathrm{CoV}_{3} \mathrm{O}_{8}$ formula.

## Magnetic Susceptibility of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$

The magnetic susceptibility $\chi$ of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ exhibits a sharp peak at 8.2 K in the $\chi-T$ curve that must be a Néel


FIG. 2. Structures of polyhedral units: (a) $\mathrm{MO}_{6}-M \mathrm{O}_{6}$ ( $M=\mathrm{Co}$ and $\mathrm{V}(1)$ ) unit sharing $\mathrm{O}(5)$ vertex; (b) $\mathrm{V}(2) \mathrm{O}_{4}-\mathrm{V}(3) \mathrm{O}_{5}$ unit.
temperature below which antimagnetic order takes place. Typical Curie-Weiss paramagnetic behavior is observed above 9 K , yielding an effective moment of $5.75 \mu_{\mathrm{B}}$ and a Weiss constant of -32.1 K . The effective moment is in good agreement with the calculated value of $5.60 \mu_{\mathrm{B}}$ for one $\mathrm{Co}^{2+}(S=3 / 2)$ and one $\mathrm{V}^{4+}(S=1 / 2)$ ion in the formula $\mathrm{CoV}_{3} \mathrm{O}_{8}$. Since $\mathrm{V}(2)$ and $\mathrm{V}(3)$ are pentavalent, Co and $\mathrm{V}(1)$ are confirmed to be di- and tetravalent, respectively. The negative Weiss constant is indicative of antiferromagnetic interaction in the $M \mathrm{O}_{6}$ slabs.

## Comparison between Ibam and Iba2

There is another choice of space group for $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$, namely, the noncentrosymmetric $I b a 2$ as a replacement for the centrosymmetric Ibam. Actually, the structure of $\alpha$ $\mathrm{ZnV}_{3} \mathrm{O}_{8}$ was determined using Iba2 instead of Ibam (4), but the reason for this choice was not described. To work out the space group problem, the $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ structure was

TABLE 3
Bond Distances ( $\AA \AA$ ) and Angles ( ${ }^{\circ}$ ) for Polyhedra in $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}{ }^{a}$

| $\mathrm{CoO}_{6}$ octahedron |  |  |  |
| :---: | :---: | :---: | :---: |
| $\mathrm{Co}-\mathrm{O}(1)$ | 1.991(3) | $\mathrm{Co}-\mathrm{O}(2)^{i}$ | 2.002(2) |
| $\mathrm{Co}-\mathrm{O}(3)^{i}$ | 2.109(3) | $\mathrm{Co}-\mathrm{O}(5)^{i i}$ | 2.02(1) |
| $\mathrm{Co}-\mathrm{O}(6)$ | 2.008(2) | $\mathrm{Co}-\mathrm{O}(6)^{\text {iii }}$ | $2.105(2)$ |
| $\mathrm{O}(1)-\mathrm{Co}-\mathrm{O}(2)^{i}$ | 94.1(2) | $\mathrm{O}(1)-\mathrm{Co}-\mathrm{O}(3)^{i}$ | 169.6(2) |
| $\mathrm{O}(1)-\mathrm{Co}-\mathrm{O}(5)^{i i}$ | 94.3(4) | $\mathrm{O}(1)-\mathrm{Co}-\mathrm{O}(6)$ | 96.8(2) |
| $\mathrm{O}(1)-\mathrm{Co}-\mathrm{O}(6)^{i i i}$ | 85.1(2) | $\mathrm{O}(2)^{i}-\mathrm{Co}-\mathrm{O}(3)^{i}$ | 76.8(2) |
| $\mathrm{O}(2)^{i}-\mathrm{Co}-\mathrm{O}(5)^{i i}$ | 111.3(3) | $\mathrm{O}(2)^{i}-\mathrm{Co}-\mathrm{O}(6)$ | 160.3(1) |
| $\mathrm{O}(2)^{i}-\mathrm{Co}-\mathrm{O}(6)^{i i i}$ | 87.9(1) | $\mathrm{O}(3)^{i}-\mathrm{Co}-\mathrm{O}(5)^{i i}$ | 93.2(4) |
| $\mathrm{O}(3)^{i}-\mathrm{Co}-\mathrm{O}(6)$ | 89.4(2) | $\mathrm{O}(3)^{i}-\mathrm{Co}-\mathrm{O}(6)^{i i i}$ | 90.6(2) |
| $\mathrm{O}(5)^{i i}-\mathrm{Co}-\mathrm{O}(6)$ | 89.2(3) | $\mathrm{O}(5)^{i i}-\mathrm{Co}-\mathrm{O}(6)^{i i i}$ | 165.8(3) |
| $\mathrm{O}(6)-\mathrm{Co}-\mathrm{O}(6)^{\text {iii }}$ | 76.8(1) |  |  |
| $\mathrm{V}(1) \mathrm{O}_{6}$ octahedron |  |  |  |
| $\mathrm{V}(1)-\mathrm{O}(1)$ | 2.000(3) | $\mathrm{V}(1)-\mathrm{O}(2)^{i}$ | 1.966 (3) |
| $\mathrm{V}(1)-\mathrm{O}(3)^{i}$ | 2.109(3) | $\mathrm{V}(1)-\mathrm{O}(5)^{i i}$ | 1.69(1) |
| $\mathrm{V}(1)-\mathrm{O}(6)$ | 2.075(3) | $\mathrm{V}(1)-\mathrm{O}(6)^{i i i}$ | 2.208 (3) |
| $\mathrm{O}(1)-\mathrm{V}(1)-\mathrm{O}(2)^{i}$ | 95.0(2) | $\mathrm{O}(1)-\mathrm{V}(1)-\mathrm{O}(3)^{i}$ | 167.0(2) |
| $\mathrm{O}(1)-\mathrm{V}(1)-\mathrm{O}(5)^{i i}$ | 98.5(4) | $\mathrm{O}(1)-\mathrm{V}(1)-\mathrm{O}(6)$ | 94.4(2) |
| $\mathrm{O}(1)-\mathrm{V}(1)-\mathrm{O}(6)^{i i i}$ | 82.2(1) | $\mathrm{O}(2)^{i}-\mathrm{V}(1)-\mathrm{O}(3)^{i}$ | 77.5(2) |
| $\mathrm{O}(2)^{i}-\mathrm{V}(1)-\mathrm{O}(5)^{i i}$ | 106.1(4) | $\mathrm{O}(2)^{i}-\mathrm{V}(1)-\mathrm{O}(6)$ | 155.7(2) |
| $\mathrm{O}(2)^{i}-\mathrm{V}(1)-\mathrm{O}(6)^{i i i}$ | 85.9(2) | $\mathrm{O}(3)^{i}-\mathrm{V}(1)-\mathrm{O}(5)^{i i}$ | 93.7(4) |
| $\mathrm{O}(3)^{i}-\mathrm{V}(1)-\mathrm{O}(6)$ | 88.8(2) | $\mathrm{O}(3)^{i}-\mathrm{V}(1)-\mathrm{O}(6)^{i i i}$ | 86.7(2) |
| $\mathrm{O}(5)^{i i}-\mathrm{V}(1)-\mathrm{O}(6)$ | 94.6(4) | $\mathrm{O}(5)^{i i}-\mathrm{V}(1)-\mathrm{O}(6)^{i i i}$ | 167.8(4) |
| $\mathrm{O}(6)-\mathrm{V}(1)-\mathrm{O}(6)^{\text {iii }}$ | 73.2(1) |  |  |
| $\mathrm{V}(2) \mathrm{O}_{6}$ tetrahedron |  |  |  |
| $\mathrm{V}(2)-\mathrm{O}(3)^{i v}$ | $1.696(3)$ | $\mathrm{V}(2)-\mathrm{O}(4)$ | 1.783(2) |
| $\mathrm{V}(2)-\mathrm{O}(6)$ | 1.701(2) | $\mathrm{V}(2)-\mathrm{O}(6)^{v}$ | 1.701(2) |
| $\mathrm{O}(3)^{i v}-\mathrm{V}(2)-\mathrm{O}(4)^{v}$ | 110.8(1) | $\mathrm{O}(3)^{i v}-\mathrm{V}(2)-\mathrm{O}(6)^{v}$ | 107.44(7) |
| $\mathrm{O}(3)^{i v}-\mathrm{V}(2)-\mathrm{O}(6)^{v i}$ | 107.44(7) | $\mathrm{O}(4)^{v}-\mathrm{V}(2)-\mathrm{O}(6)^{v}$ | 111.50(7) |
| $\mathrm{O}(4)^{v}-\mathrm{V}(2)-\mathrm{O}(6)^{v i}$ | 111.50(7) | $\mathrm{O}(6)^{v}-\mathrm{V}(2)-\mathrm{O}(6)^{v i}$ | 108.0(1) |
| $\mathrm{V}(3) \mathrm{O}_{5}$ trigonal bipyramid |  |  |  |
| $\mathrm{V}(3)-\mathrm{O}(1)$ | 1.650(2) | $\mathrm{V}(3)-\mathrm{O}(1)^{v}$ | 1.650(2) |
| $\mathrm{V}(3)-\mathrm{O}(2)^{v i}$ | 1.854(3) | $\mathrm{V}(3)-\mathrm{O}(4)^{i v}$ | 2.082(3) |
| $\mathrm{V}(3)-\mathrm{O}(4)^{v i i}$ | 1.997(3) |  |  |
| $\mathrm{O}(1)-\mathrm{V}(3)-\mathrm{O}(1)^{v}$ | 106.3(1) | $\mathrm{O}(1)-\mathrm{V}(3)-\mathrm{O}(2)^{v i}$ | 101.26(8) |
| $\mathrm{O}(1)-\mathrm{V}(3)-\mathrm{O}(4)^{\text {iv }}$ | 126.19(7) | $\mathrm{O}(1)-\mathrm{V}(3)-\mathrm{O}(4)^{v i i}$ | 96.48(8) |
| $\mathrm{O}(1)^{v}-\mathrm{V}(3)-\mathrm{O}(2)^{v i}$ | 101.26(8) | $\mathrm{O}(1)^{v}-\mathrm{V}(3)-\mathrm{O}(4)^{i v}$ | 126.18(7) |
| $\mathrm{O}(1)^{v}-\mathrm{V}(3)-\mathrm{O}(4)^{v i i}$ | 96.48(8) | $\mathrm{O}(2)^{v i}-\mathrm{V}(3)-\mathrm{O}(4)^{i v}$ | 80.9(1) |
| $\mathrm{O}(2)^{v i}-\mathrm{V}(3)-\mathrm{O}(4)^{v i i}$ | 150.1(1) | $\mathrm{O}(4)^{i v}-\mathrm{V}(3)-\mathrm{O}(4)^{v i i}$ | 69.2(1) |

${ }^{a}$ Symmetry codes: $i, x-1 / 2, y+1 / 2, z+1 / 2 ; i i, x, y+1, z ; i i i, 1 / 2-x$, $3 / 2-y, 1 / 2-z ; i v, x-1 / 2,3 / 2-y, z ; v, x, y,-z ; v i, 1 / 2-x, x, 1 / 2+y, z ; v i i$, $1 / 2-x, y-1 / 2, z$.
modified using the space group $I b a 2$ by placing all the atoms in $8 c$ positions. The structure was refined to $R=0.034$ and $R_{\mathrm{w}}=0.041$ for 110 variables, which are compared with $R=0.034$ and $R_{\mathrm{w}}=0.030$ for 67 variables for the space group Ibam. The structural difference is seen in the atomic positions of the $\mathrm{MO}_{6}$ slab as depicted in Fig. 4. For the space group Ibam (Fig. 4a), Co, V(1), and O(5) sites are half-occupied whereas for $\operatorname{Iba2}$ (Fig. 4b) their sites are clearly separated and arranged in an ordered array. The bond distances for $I b a 2$ become $\mathrm{Co}-\mathrm{O}(5)=1.97(1) \AA$ and $\mathrm{V}(1)-\mathrm{O}(5)=1.75(1) \AA$, which give BVS values 2.55 , 3.31 , and 1.54 for $\mathrm{Co}, \mathrm{V}(1)$, and $\mathrm{O}(5)$, respectively, being even worse


FIG. 3. $M \mathrm{O}_{6}$ slab standing in the $b c$ plane: (a) polyhedral representation showing linkages of $M \mathrm{O}_{6}$ chains through $\mathrm{O}(5)$ atoms denoted by shaded circles; (b) an example of the atomic arrangement of $\mathrm{Co}, \mathrm{V}(1)$, and $\mathrm{O}(5)$ atoms denoted by small open, small shaded, and large shaded circles, respectively.
than those for Ibam. Moreover, the site occupancies of Co and $\mathrm{V}(1)$, fixed to unity in the foregoing calculations, were refined to $0.93(1)$ and $1.07(2)$, respectively, both of which correspond to $25 \mathrm{e} /$ atom, namely, the mean value of electron numbers of Co and V . This indicates that half of the Co site

TABLE 4
Bond Valence Sums for Individual Atoms of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{\mathbf{8}}$

|  | Co | $\mathrm{V}(1)$ | $\mathrm{V}(2)$ | $\mathrm{V}(3)$ | Total |
| :--- | :--- | :--- | :--- | :--- | :--- |
| $\mathrm{O}(1)$ | 0.45 | 0.56 |  | $1.51,1.51$ | $2.02^{a}$ |
| $\mathrm{O}(2)$ | 0.43 | 0.61 |  | 0.87 | 1.91 |
| $\mathrm{O}(3)$ | 0.32 | 0.42 | 1.34 |  | 2.08 |
| $\mathrm{O}(4)$ |  |  | 1.06 | $0.47,0.59$ | 2.12 |
| $\mathrm{O}(5)$ | 0.41 | 1.29 |  |  | 1.70 |
| $\mathrm{O}(6)$ | $0.43,0.33$ | $0.46,0.32$ | $1.32,1.32$ |  | $2.08^{b}$ |
| Total | 2.37 | 3.66 | 5.04 | 4.96 |  |

${ }^{a}(0.45+0.56) / 2+1.51$.
${ }^{b}(0.43+0.33) / 2+(0.46+0.32) / 2+1.32$.
is occupied by a V atom and half of the $\mathrm{V}(1)$ site is occupied by a Co atom just as in the case of Ibam. Consequently, judging from the reliability factors, the BVS values, and the site occupancies, the space group Iba2 should be ruled out.

## Structural Relation between the $\alpha$ and $\beta$ forms

It was reported that $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ transforms to $\beta-\mathrm{CoV}_{3} \mathrm{O}_{8}$ at $650^{\circ} \mathrm{C}$, which crystallizes in the $C$-centered monoclinic system (2). Unfortunately, we failed to obtain single crystals of the $\beta$ form by heating $\alpha$-form crystals in an evacuated silica tube. The $\beta$ phase also exists in $A=\mathrm{Mg}$ and $\mathrm{Zn}(2,3)$, of which $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$ has been structurally characterized by Saux and Galy (5): $C 2 / m$ with $a=10.293(9) \AA, b=$ $8.530(8) \AA, c=7.744(8) \AA, \beta=119.5(5)^{\circ}$, and $Z=4$. It is quite likely that the structure of $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$ is the same as that of $\beta-\mathrm{CoV}_{3} \mathrm{O}_{8}$ and then we can compare the $\alpha$ and $\beta$ forms. The structure of the $\beta$ form consists of $\mathrm{MO}_{6}$ octahedra ( $M=\mathrm{Mg}, \mathrm{V}$ ) and $\mathrm{VO}_{4}$ tetrahedra. The octahedral $M$ site is occupied equally by Mg and V atoms, which were refined using the same atomic parameters (5). Furthermore, $M \mathrm{O}_{6}$ chains are linked by sharing vertices of $\mathrm{O}(1)$ to form an $M \mathrm{O}_{6}$ slab which stands in $(10 \overline{2})$ as shown in Fig. 5. The $M \mathrm{O}_{6}$ slab of the $\beta$ form is structurally similar to that of the $\alpha$ form (Fig. 3a) but is different in the locations of shared vertices. The $M \mathrm{O}_{6}$ slabs of the $\beta$ form are bridged through a $\mathrm{V}_{2} \mathrm{O}_{7}$ or $\mathrm{VO}_{4}-\mathrm{VO}_{4}$ unit made of two $\mathrm{VO}_{4}$ tetrahedra by sharing vertices. This $\mathrm{VO}_{4}-\mathrm{VO}_{4}$ unit is compared to the $\mathrm{V}(2) \mathrm{O}_{4}-\mathrm{V}(3) \mathrm{O}_{5}$ unit in the $\alpha$ form. According to Saux and Galy (5), the $M$ site of the $\beta$ form is statistically occupied by $\mathrm{Mg}^{2+}$ and $\mathrm{V}^{4+}$ ions, but no further discussion was given. Judging from the structural analogy of the $M \mathrm{O}_{6}$ slabs between the $\alpha$ and $\beta$ forms, we believe that the Mg and V atoms are distributed so that the $\mathrm{O}(1)$ atom of the shared vertex bonds to the Mg and V atoms on opposite sides in the same manner as the Co and V atoms in the $\alpha$ form. If this is true, the $\mathrm{O}(1)$ atom in the $4 g$ position $(0, y, 0)$ must shift to the side of the V atom. This can be realized by converting

b $\quad$ lba2


FIG. 4. Comparison of arrangement of atom sites in the $M \mathrm{O}_{6}$ slabs between the space groups (a) Ibam and (b) Iba2. Atom sites of Co, V(1), and $\mathrm{O}(5)$ are denoted by small open, small shaded, and large shaded circles, respectively.


FIG. 5. $M \mathrm{O}_{6}$ slab of $\beta$-form crystal standing in (10 $\left.\overline{2}\right)$ for $\beta-\mathrm{MgV}_{3} \mathrm{O}_{8}$ (Ref. 5). Shaded circles denote $\mathrm{O}(1)$ atoms of shared vertices of neighboring $M \mathrm{O}_{6}$ chains.
the $4 g$ position to a half-occupied $8 j$ position $(x, y, z)$ with parameters of $x=-\delta_{1}$ and $z=-\delta_{2}$ just like the positional change of the $\mathrm{O}(5)$ atom of the $\alpha$ form from $8 f(x, 0,1 / 4)$ to $16 k\left(x, \delta_{1}, \frac{1}{4}-\delta_{2}\right)$. In conclusion, the structures of the $\alpha$ and $\beta$ forms are analogous, particularly in the construction of $M \mathrm{O}_{6}$ slabs.

## CONCLUSION

The structure of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ has been determined based on the space group Ibam, which has been proved to be more appropriate than the space group Iba 2 . The structure consists of $M \mathrm{O}_{6}(M=\mathrm{Co}, \mathrm{V})$ octahedral slabs and $\mathrm{VO}_{4}-\mathrm{VO}_{5}$ units and the slabs are made of edge-sharing $\mathrm{MO}_{6}$ zigzag chains. $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ is a typical class I mixed-valence $\left(\mathrm{V}^{\mathrm{IV} / \mathrm{V}}\right)$ compound (9), since $\mathrm{V}^{4+}$ ions occupy octahedral sites and $\mathrm{V}^{5+}$ ions occupy both tetraherdal and trigonal-bipyramidal sites. Related class I mixed-valence compounds are found in $\mathrm{BaV}_{3} \mathrm{O}_{8}$ (10) with $\mathrm{V}^{\mathrm{IV}} \mathrm{O}_{6}$ octahedra and $\mathrm{V}^{\mathrm{V}} \mathrm{O}_{4}$ tetrahedra and $A_{2} \mathrm{~V}_{3} \mathrm{O}_{8}\left(A=\mathrm{K}(11), \mathrm{Rb}\right.$ (12), $\mathrm{NH}_{4}$ (13)) with $\mathrm{V}^{\mathrm{IV}} \mathrm{O}_{5}$ square pyramids and $\mathrm{V}^{\mathrm{V}} \mathrm{O}_{4}$ tetrahedra. A striking structural feature of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ is seen in the metal distribution over the $M$ site in which Co and V atoms equally reside. The metal distribution is basically random but is constrained by the rule that the two metals on opposite sides of the vertex
shared by neighboring $\mathrm{MO}_{6}$ chains must be Co and V . This rule guarantees the equal occupations of Co and V , in other words the stoichiometric formula $\mathrm{CoV}_{3} \mathrm{O}_{8}$. The magnetic susceptibility data suggest some antiferromagnetic order in the $M \mathrm{O}_{6}$ slabs.

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